



Numerical modelling of concrete curing, regarding hydration and temperature phenomena

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Abstract

A numerical model that accounts for the hydration and aging phenomena during the early ages of concrete curing is presented in a format suitable for a finite element implementation. Assuming the percolation of water through the hydrates already formed as the dominant mechanism of cement hydration, the model adopts an internal variable called hydration degree, whose evolution law is easily calibrated and allows an accurate prediction of the hydration heat production. Compressive strength evolution is related to the aging degree, a concept that accounts for the influences of the hydration and curing temperature on the final mechanical properties of concrete. The model capabilities are illustrated by means of a wide set of experimental tests involving ordinary and high performance concretes, and through the simulation of the concrete curing on a viaduct deck of the Öresund Link.

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1. Introduction

A rising tendency to use concrete as a structural material at early ages, that is, before completion of the hydration process, is observed nowadays. This is evident for pre-casting as well as for pre-stressed concrete, where anticipation of the loading of concrete provides economical benefits concerned with the reusing of formworks, as well as abbreviating the construction phase. Another field where high cadence in production is registered concerns rolled compacted concrete dams, where construction technologies resemble the ones usually adopted for roads. The quality control inherent to these situations is a subject of major concern, since durability and functionality of the structures is greatly affected by

undesirable damages like cracking, at early ages primarily associated with thermal and shrinkage effects.

A domain of research to which the present paper is mostly concerned refers to the modelling of such phenomena in a format suitable for structural applications, namely by providing a mathematical representation of the observed temperature and volume changes that take place in concrete associated to the hydration of cement, as well as the increase of strength and stiffness of the hardening paste due to aging. The model to be presented herein accounts for many of the relevant features of concrete behaviour at early ages, but in order to make it appropriate for an implementation within the context of the finite element method, the kinetics of the chemical reaction will be viewed on a macroscopic level, that is, at the scale of the specimens tested on laboratory.

Basically such approach corresponds to visualise the enhancement of the mechanical properties of concrete as due to changes in the concentration of the many

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constituents existing at the microscale, rather than an actual change of the mechanical properties of such components. Within this framework the aging process is therefore linked to the formation of new hydrates, and consequently the essential phenomenon which dominates the hydration kinetics is the diffusion of the free water between the layers of the already formed hydrates [1–3], here reproduced via a normalised variable termed ‘hydration degree’. Its evolution obviously depends on the current value of the hydration degree, but owing to the thermally activated nature of cement hydration a strong dependency on the temperature is also observed, a feature usually accomplished by assuming an Arrhenius type kinematics.

Additionally an aging model is proposed to describe the compressive strength enhancement during the hydration process. This is based on the novel concept of ‘aging degree’, with a rate evolution here linked both with the hydration degree and with the curing temperature. Like in models purely based on maturity concepts this accounts for the influence of the temperature on the compressive strength evolution, but additionally it allows to capture the effect of the curing temperature history on the final concrete strengths.

The model parameters involved are easily obtainable on the basis of adiabatic or isothermal tests. For the validation of the proposed model a wide set of experimental tests concerning the temperature evolution and the rising of the compressive strength of several concrete mixes under controlled casting temperatures is presented. Besides, it is demonstrated that the proposed model performs quite well both for ordinary concrete and for high performance concrete with silica fume admixtures.

The article closes with a structural application, concerned with the simulation of the curing of a concrete deck of one of the viaducts of the Öresund Link (between Denmark and Sweden). Since during the concrete casting the selected viaduct was extensively monitored, comparison of the experimentally registered temperatures with the ones obtained via a numerical implementation of the proposed formulation will provide a fruitful discussion concerning the capabilities of the model here presented.

2. Hydration of concrete

2.1. Thermo-chemical model

The chemical processes associated to the hardening of concrete in the first days after casting are accompanied by significant temperature changes, since cement hydration is a highly exothermic and thermally activated reaction. Based on the theory of reactive porous media developed by Coussy in [1], as well as on its application

to concrete under the macroscopic approach proposed by Ulm and Coussy in [2,3], the thermo-chemical model to be presented next is based on the macroscopic interpretation of the cement hydration as a chemical reaction in which the free water is a reactant phase, which combines with the unhydrated cement to form hydrates as a product phase.

Since the dominant mechanism in the kinetics of the reaction is due to the dispersion of water, if one defines χ as the number of moles of water combined per unit volume, this entity can be considered an internal variable of the system. For a more suitable handling and interpretation let us define this internal variable in the normalised format

$$\xi = \frac{\chi}{\bar{\chi}_{\infty}} \quad (1)$$

where $\bar{\chi}_{\infty}$ is the ideal final value of χ (attained in perfect conditions of the water/cement ratio w/c and other factors ensuring full hydration) and ξ is what one could call the hydration degree, an indication of how much the hydration reaction has already advanced. In practical conditions the full hydration of the cement is not obtained; consequently $\chi_{\infty} < \bar{\chi}_{\infty}$ and the hydration degree at infinite time is such that $\xi_{\infty} < 1$.¹

Following [2] the rate $\dot{\xi}$ is itself a function of the hydration degree, but since the combination of water with cement is a thermally activated reaction, an amplification based on the widely used Arrhenius type law will be also considered to account for the influence of the temperature [5]. A positive evolution of the form

$$\dot{\xi} = \tilde{A}(\xi) \exp\left(-\frac{E_a}{RT}\right) \geq 0 \quad (2)$$

is therefore proposed [2,3], where apart from the temperature T the Arrhenius exponential term incorporates the constant R for the ideal gases and the activation energy of the reaction E_a . Eq. (2) encompasses the direct influence of the actual value of the hydration degree on term \tilde{A} , a normalised affinity to be considered specific for each hydration reaction, and easily obtainable from an adiabatic calorimetric test, as it will be clarified later. In the definition of the normalised chemical affinity it is convenient to express it as

$$\tilde{A} = A\eta \quad (3)$$

where A is the chemical affinity (strictly speaking) and η expresses the permeability directly linked with the microcirculation of the free water through the hydrates. By

¹ By relating ξ_{∞} with w/c in [4] it is suggested that the formula $\xi_{\infty} = 1.031(w/c)/(0.194 + w/c)$ could be used for an estimation of the final hydration degree.

assuming that the percolation of the water is retarded with the growth of the hydrates, η would have to decrease with ξ , and consequently induced by [6] it is here postulated that

$$\eta = \frac{1}{n_0} \exp\left(-\frac{\bar{n}\xi}{\xi_\infty}\right) \quad (4)$$

where n_0 and \bar{n} are material constants.

Now, by defining the chemical affinity with a format like [7]

$$A = k\left(\frac{A_0}{k\xi_\infty} + \xi\right)(\xi_\infty - \xi) \quad (5)$$

where A_0 is the initial affinity of the reaction (that is, when $\xi = 0$) and k is an additional parameter, handling Eqs. (2)–(5) the final form for $\dot{\xi}$ reads

$$\dot{\xi} = \frac{k}{n_0} \left(\frac{A_0}{k\xi_\infty} + \xi\right)(\xi_\infty - \xi) \exp\left(-\frac{\bar{n}\xi}{\xi_\infty}\right) \exp\left(-\frac{E_a}{RT}\right) \geq 0 \quad (6)$$

From the observation of this rate equation it becomes evident that the thermal activation expressed by the Arrhenius factor is of primary importance at the beginning of the reaction, but as the hydration proceeds considerable attenuation is introduced by the permeability exponential term as well as by factor $(\xi_\infty - \xi)$, cancelling the evolution as the hydration degree reaches the final value $\xi = \xi_\infty$. Therefore, and beyond the required positiveness for $\dot{\xi}$, the evolution expressed in Eq. (6) contains the basic requisites for a realistic modelling of the hydration reaction.

2.2. Thermal field equation

2.2.1. Thermo-dynamic equilibrium

According to the first and second principles of thermodynamics a domain undergoing thermal changes is ruled by a field equation of the form

$$C\dot{T} = \dot{Q} + \dot{H} - \nabla \cdot \mathbf{Q}_f + \dot{D} \quad (7)$$

where C denotes the heat capacity per unit volume, Q is the heat associated to an internal source, H is the external heat source, Q_f is the heat flux and D is the dissipation. Focusing on the simulation of the hydration problem it will be assumed that C is constant, the dissipation is taken as negligible, the same occurring with the external heat source. Furthermore, the internal heat source Q is provided by the hydration reaction itself, and according to many authors [6,8–10] hereafter it will be assumed that $\dot{Q} = Q_\xi \dot{\xi}$, with Q_ξ being a material constant. Therefore Eq. (7) turns into

$$C\dot{T} = Q_\xi \dot{\xi} + k_T \nabla \cdot (\nabla T) \quad (8)$$

where $Q_f = -k_T \nabla T$ was assumed according to the Fourier's law (k_T designates the thermal conductivity). Note that due to the dependency of ξ on T the term due to the hydration heat \dot{Q} acts as a nonlinear internal heat source, and accordingly an iterative procedure is usually required to solve Eq. (8) in space and time for general situations, as it will be discussed on the forthcoming section.

Besides the thermal field equilibrium expressed by Eq. (8), the Dirichlet and Newman boundary conditions have to be introduced. The first one concerns the temperature \bar{T} prescribed along boundary Γ_T , whereas the second has to do with condition $Q_f \cdot \mathbf{n} = \bar{q}$, expressing the heat flux along the normal \mathbf{n} to boundary Γ_q . In the applications to be presented in this paper \bar{q} can be identified with the heat flux due to convection–radiation phenomena (if any), and accordingly one takes $\bar{q} = h(T - T_{\text{ext}})$, with h being the convection–radiation coefficient and T_{ext} denoting the external temperature.

2.2.2. Finite element implementation

Following standard reasoning and assuming the usual interpolation strategy $T = \mathbf{N}T^e$, where \mathbf{N} denotes the interpolation matrix and T^e designates the nodal temperatures for a given finite element with volume Ω , the weak form for the thermal equations (accounting for the Newman boundary conditions as well) becomes

$$\int_\Omega \mathbf{N}^T C \dot{T} d\Omega + \int_\Omega \nabla \mathbf{N}^T k_T \nabla T d\Omega = \int_\Omega \mathbf{N}^T Q_\xi \dot{\xi} d\Omega - \int_{\Gamma_q} \mathbf{N}^T \bar{q} d\Gamma_q \quad (9)$$

(if \mathbf{N} trivially fulfils the Dirichlet conditions). Writing this equation for time t_{n+1} and assuming a backward-Euler integration scheme of the form

$$\dot{T}_{n+1} = \frac{(T_{n+1} - T_n)}{\Delta t} \quad (10a)$$

$$\dot{\xi}_{n+1} = \frac{(\xi_{n+1} - \xi_n)}{\Delta t} \quad (10b)$$

with Δt being the time-step between two consecutive instants t_n and t_{n+1} , Eq. (9) leads to the following equation, suitable for computational implementation:

$$\frac{1}{\Delta t} \mathbf{C}^e (T_{n+1}^e - T_n^e) + \mathbf{K}^e T_{n+1}^e = \mathbf{F}_T^e + \mathbf{F}_\xi^e \quad (11)$$

Following Eqs. (9)–(10b) the elemental matrices and ‘force vectors’ involved in this equation are computed according to

$$\mathbf{C}^e = \int_\Omega \mathbf{N}^T C N d\Omega \quad (12a)$$

$$\mathbf{K}^e = \int_{\Omega} \nabla \mathbf{N}^T k_T \nabla \mathbf{N} d\Omega + \int_{\Gamma_q} \mathbf{N}^T h \mathbf{N} d\Gamma_q \quad (12b)$$

$$\mathbf{F}_T^e = \int_{\Gamma_q} \mathbf{N}^T h T_{\text{ext}} d\Gamma_q \quad (12c)$$

$$\mathbf{F}_{\xi}^e = \int_{\Omega} \mathbf{N}^T Q_{\xi} \dot{\xi}_{n+1} d\Omega \quad (12d)$$

which through the using of standard assembling procedures will provide the global \mathbf{C} and \mathbf{K} matrices and the \mathbf{F}_T and \mathbf{F}_{ξ} vectors required for the finite element analysis of a structural problem, that is,

$$\left(\frac{\mathbf{C}}{\Delta t} + \mathbf{K} \right) \mathbf{T}_{n+1} = \mathbf{F}_T + \mathbf{F}_{\xi} + \frac{\mathbf{C}}{\Delta t} \mathbf{T}_n \quad (13a)$$

Due to the dependency of the hydration rate on the temperature, expressed in Eq. (6), a two-level iterative procedure needs to be implemented to solve Eq. (13a): (i) at a structural level, due to the nonlinear dependency of \mathbf{F}_{ξ} on \mathbf{T}_{n+1} ; (ii) at a local level (that is, at each integration point), due to the nonlinear dependency of the internal variable ξ on the free variable T . By recalling the backward-Euler scheme reproduced in Eq. (10b) the following discrete form could be attributed to Eq. (6):

$$\begin{aligned} \xi_{n+1} - \xi_n - \frac{\Delta t k}{n_0} \left(\frac{A_0}{k \xi_{\infty}} + \xi_{n+1} \right) (\xi_{\infty} - \xi_{n+1}) \\ \times \exp \left(-\frac{\bar{n} \xi_{n+1}}{\xi_{\infty}} \right) \exp \left(-\frac{E_a}{RT_{n+1}} \right) = 0 \end{aligned} \quad (13b)$$

This equation is solved via the Newton–Raphson method, implemented at each integration point to extract the hydration degree ξ_{n+1} associated to the local temperature T_{n+1} . This local temperature is obtained by performing a standard interpolation over the nodal temperatures \mathbf{T}_{n+1} , the latter provided by an ‘outer’ Newton–Raphson algorithm implemented to solve the nonlinear problem expressed at the structural level by Eq. (13a). After solving Eq. (13b) for ξ_{n+1} the hydration rate $\dot{\xi}_{n+1}$ is computed from Eq. (10b), which according to Eq. (12d) provides a new trial vector for \mathbf{F}_{ξ} , and consequently an improved solution for \mathbf{T}_{n+1} via Eq. (13a).

A remark is made to the fact that, since the rate evolution of ξ is usually considerably greater than \dot{T} , the numerical integration of the hydration degree should be performed by using a time-step significantly lower than the one adopted for the structural problem [11]. Within the sub-incremental time-steps introduced to solve Eq. (13b) an interpolation is obviously required to define the sub-incremental values of the temperature at the integration point: the adoption of a linear interpolation between T_n and T_{n+1} is consistent with the time integra-

tion scheme reproduced in Eq. (13b), and consequently it was assumed for the applications of the present paper.

2.3. Experimental identification of the model parameters

Under adiabatic conditions Eq. (8) becomes

$$C \dot{T} = Q_{\xi} \dot{\xi} \quad (14)$$

Denoting by T_0 the initial temperature this equation is easily solved to render

$$T = T_0 + Q_{\xi} \xi / C \quad (15)$$

At the end of an adiabatic test $T = T_{\infty}$ and $\xi = \xi_{\infty}$, and consequently Eq. (15) can be used to evaluate the average value of constant Q_{ξ} , that is,

$$Q_{\xi} = C(T_{\infty} - T_0) / \xi_{\infty} \quad (16)$$

Furthermore, the normalised affinity \tilde{A} , an essential entity of the hydration model, is also directly ‘measurable’ during the adiabatic test. By substituting Eq. (2) into Eq. (14) owing to Eq. (16) one obtains

$$\tilde{A}_{\text{test}} = \frac{\xi_{\infty} \dot{T}}{T_{\infty} - T_0} \exp \left(\frac{E_a}{RT} \right) \quad (17)$$

This equation allows computing the normalised affinity by measuring the temperature rate \dot{T} (or its discrete equivalent $\Delta T / \Delta t$) along the experiment, together with the adiabatic temperature itself. Therefore, the three parameters k/n_0 , A_0/k and \bar{n} involved in Eq. (6) can be adjusted by fitting the model affinity \tilde{A} to the experimental one \tilde{A}_{test} obtained from an adiabatic calorimetric test according to Eq. (17).

As for the ratio E_a/R it can be experimentally measured as well, ranging within the interval 3000–8000 °K for concrete. According to [12,13] the activation energy remains almost constant above 20 °C, and consequently we will assume E_a/R as a constant, tuned in order to match the temperature rise under an adiabatic calorimetric test.

2.4. Application to ordinary and high performance concretes

Insight on the ability of the proposed model to reproduce the hydration reaction is provided in Fig. 1, where the adiabatic tests conducted by Bentz et al. [14] for an ordinary Portland concrete (OPC) and for a high performance one (HPC) are reproduced. Table 1 reports the values adopted for the various parameters involved in the numerical model. In this figure, as well as in the forthcoming ones, dots correspond to the experimental values, whereas solid lines indicate the predictions from the above described model. From observation of Fig. 1a it becomes evident that for the OPC with a ratio $w/c = 0.35$ the experimentally measured temperature

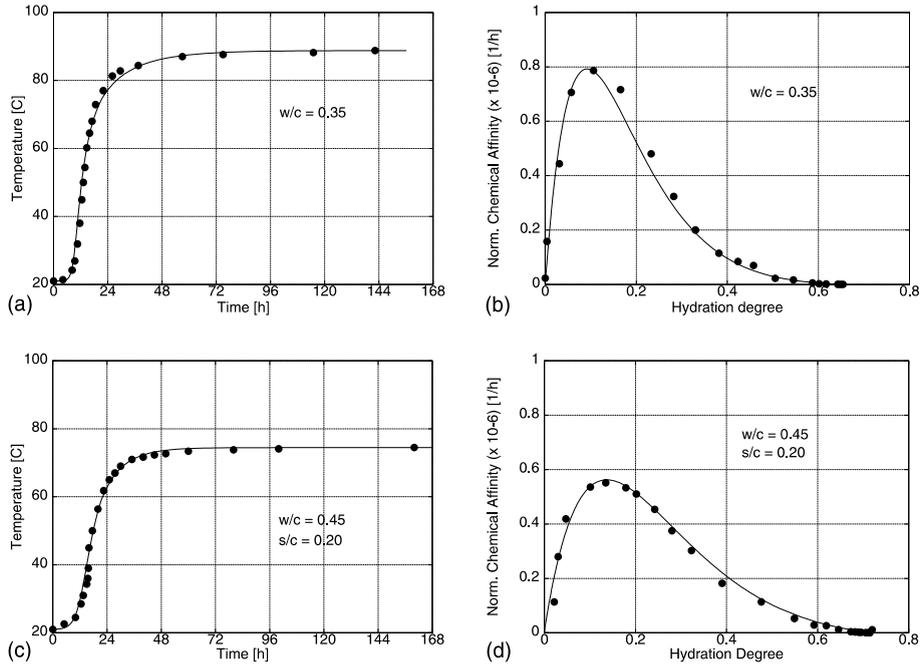


Fig. 1. Temperature evolution and normalised affinity for Bentz et al. adiabatic tests ((•••) experimental values [14], (—) model predictions).

Table 1
Material properties for Bentz et al. tests

	ξ_∞	k/n_0 (10^8 h^{-1})	\bar{n}	A_0/k (10^{-5})	E_a/R (°K)	Q_ξ (10^8 J/m^3)
OPC	0.66	0.35	6.0	1.0	5000	2.39
HPC	0.72	0.15	4.0	0.5	5000	1.73

$C = 2.33 \times 10^6 \text{ J/m}^3 \text{ }^\circ\text{C}$, $T_0 = 21 \text{ }^\circ\text{C}$.

rise is accurately predicted by the numerical model. The same applies for Fig. 1b, where the normalised affinity adopted in the model is compared with the inferred experimental one. Figs. 1c and d provide similar comparisons for the high performance concrete, where $w/c = 0.45$ and a silica fume/cement ratio $s/c = 0.2$ were considered. Good agreement is therefore obtained for both concretes.

3. Aging model and temperature effects

3.1. Hydration degree and maturity

From a phenomenological standpoint the development of the mechanical properties of concrete at early ages has to be linked with the hydration degree, as this entity may be viewed as an indicator of how much the inherent chemical reaction has already advanced. However, much experimental data put into evidence that

temperature plays also a significant role in the development of concrete strength during the curing process. To account for this influence of temperature T many aging models have been defined in terms of ‘maturity’ or ‘equivalent age’ concepts, basically through the definition of an artificial age for the concrete, which becomes superior or inferior to the real one depending on if the actual temperature history has accelerated or retarded the aging process (in comparison with what would occur under a reference temperature T_r , typically $20 \text{ }^\circ\text{C}$). A possible definition for the equivalent age t_e is the one from Freiesleben and Perdersen, revisited in [15],

$$t_e = \sum_i \exp \left[\frac{E_a}{R} \left(\frac{1}{T_r + 273} - \frac{1}{T_i + 273} \right) \right] \Delta t \quad (18)$$

Apart from the fact that the kinetics of the hydration degree is not directly visible on this definition, the assumed accumulative influence of the temperature on the curing process is such that different thermal curing conditions could result in the same maturity, irrespective to the temperature history.

However, as demonstrated in [15] and referred also in [16,17], much experimental observation puts into evidence that concrete strength is affected not only by the accumulative temperature history but also by the history itself. Typically, if a temporary rise on the curing temperature is produced during the first day after casting a

noticeable acceleration of the hydration process is observed, effect however progressively reduced if the temperature rise is produced at later ages. Conversely, temporary drops on the curing temperature lead to a retarding effect in the hydration reaction, but the influence is also attenuated as the temperature reduction is produced later.

A complementary effect of temperature concerns with the fact that a rise in the early curing temperature improves the early concrete strength, but it reduces its final strength at later ages. On the other hand, through the lowering of the curing temperature at the first ages a retarding of the hydration kinetics is observed, which produces a decay on the initial mechanical properties of concrete, eventually with a subsidiary rise on the limit strength owing to the greater uniformity on the distribution of the hydration products and to the reduced porosity [16,17]. These reasoning put into evidence the limited ‘memory’ proficiency of traditional formulae like the one reproduced in Eq. (18), since concretes cured under different temperature histories could have the same maturity and consequently the same final strength, an assumption in contradiction with the experimental evidence. Therefore a realistic aging model should account for the influence of the hydration degree, but the effect of the curing temperature would have to be explicitly included as well.

Focusing in the concrete compressive strength, let us introduce an internal variable κ such that

$$f(\kappa) = \kappa f_\infty \tag{19}$$

where f stands for the current strength and f_∞ denotes the final compressive strength at the reference temperature. As defined here variable κ corresponds to a normalised strength, hereinafter referred to as an aging degree. Since we intend to account for both the hydration degree and the temperature effects on the evolution of the compressive strength of concrete, let us postulate a rate equation for the aging degree of the form

$$\dot{\kappa} = \lambda_T(T) \lambda_\xi(\xi) \dot{\xi} \tag{20}$$

where $\lambda_T \lambda_\xi \geq 0$. Factor λ_ξ is supposed here to obey to a linear expression of the form

$$\lambda_\xi = A\xi + B \tag{21}$$

where A and B are material parameters. Factor λ_T accounts for the influence of temperature, here reproduced by the expression

$$\lambda_T = \left(\frac{100 - T}{100 - T_r} \right)^a \tag{22}$$

where T_r is the reference temperature for the determination of f_∞ (typically 20 °C) and the exponent a is a model parameter that serves to regulate the thermal incidence on the evolution of the aging degree. Note that for $T > T_r$ term λ_T becomes smaller than one, thus attenuating the strength increase; conversely for $T < T_r$ it results $\lambda_T > 1$, thus enhancing the evolution of the aging degree. Fig. 2 illustrates the typical evolutions obtained for ξ and f when curing takes place under three different isothermal conditions: 5, 20 and 40 °C. Note in Fig. 2a that the final value for the hydration degree is the same for the three curing temperatures, and consequently the maturity and equivalent age models can only reproduce the ascending branch of the compressive strength evolution reproduced in Fig. 2b, not the change on the final plateau.

3.2. Application to Kim et al. tests

This section is dedicated to the simulation of the experimental tests reported by Kim et al. in [15], which were performed to evaluate the influence of temperature on the evolution of the compressive strength of concrete at early ages, as well as on its final values. The curing was carried out under controlled isothermal conditions during the first 28 days, namely at the temperatures of 5, 20 and 40 °C, as depicted on the first three shadowed rows of Fig. 3. Furthermore, a complex experimental program was pursued around the curing temperature of 20 °C, consisting in raising it up to 40 °C or lowering it until 5 °C during just one day, and returning to the

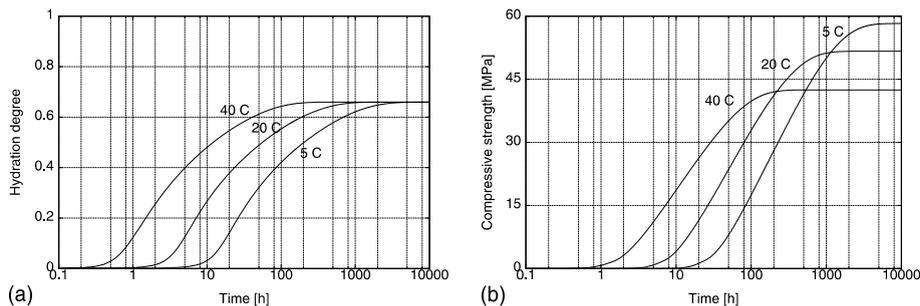


Fig. 2. Influence of temperature on the hydration and aging phenomena.

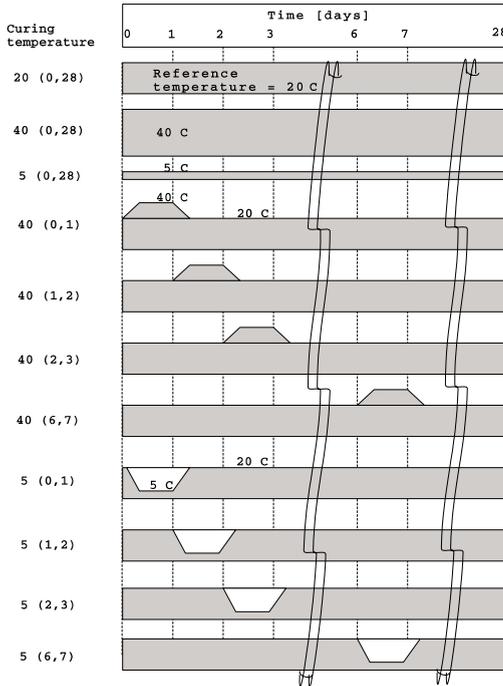


Fig. 3. Curing temperature histories in Kim et al. tests.

reference temperature of 20 °C during the remaining 27 days. Of great relevance is the fact that the temperature perturbation is produced at different ages, as also depicted in Fig. 3. To ensure a clear identification of the different curing conditions for these tests we will retain the notation adopted in the above cited reference:

40(*d*, *d* + 1) point out that a raise from 20 to 40 °C is produced between days *d* and *d* + 1, whereas 5(*d*, *d* + 1) indicate a decay from 20 to 5 °C during the same period (see also Fig. 3 for clarification). It is here emphasised that in spite of the temporary variations introduced in the curing temperature these tests can be considered as ‘stepwise’ isothermal. Two different w/c ratios of 0.35 and 0.55 were considered for the above referred tests, reproduced by the numerical model with the material properties described in Table 2.

Focusing on the first three curing conditions of Fig. 3, Figs. 4a and b allow to compare the compressive strength evolutions predicted by the numerical model against the ones obtained during the experimental tests, for the two different w/c ratios. The overall agreement of the aging model predictions with the test results is fairly good, since it captures the double influence of the curing temperature: (i) as an activator of the hydration reaction and (ii) as a conditioning factor of the final concrete strength.

The next figures concern the remaining tests sketched in Fig. 3: Fig. 5 is dedicated to the 40(0, 1), 40(1, 2), 40(2, 3) and 40(6, 7) tests, whereas Fig. 6 reproduces the 5(0, 1), 5(1, 2), 5(2, 3) and 5(6, 7) ones, all of them plotted in parallel with the 20(0, 28) curve to highlight the differences attributable to the one day variations on the isothermal 20 °C curing temperature. In order to explore the capabilities of the proposed model on different w/c ratios Fig. 5 refers to w/c = 0.35, whereas Fig. 6 corresponds to w/c = 0.55. Predictions from the aging model compare remarkably well with the experimental observations for the whole set of tests, namely in what concerns: (i) the accelerating effect introduced on the

Table 2
Material properties for Kim et al. tests

w/c	ξ_∞	k/n_0 (10^{10} h^{-1})	\bar{n}	A_0/k (10^{-5})	E_a/R (°K)	Q_c (10^8 J/m^3)	<i>A</i>	<i>B</i>	ξ_{set}	f_∞ (MPa)	<i>a</i>
0.35	0.66	4.33	5.8	1.0	7000	1.25	3.79	-1.17	0.2	52	0.7
0.55	0.76	1.20	7.5	0.1	6500	1.25	3.72	-1.85	0.3	42	0.9

$C = 2.07 \times 10^6 \text{ J/m}^3 \text{ }^\circ\text{C}$, $T_0 = 21 \text{ }^\circ\text{C}$, $T_r = 20 \text{ }^\circ\text{C}$.

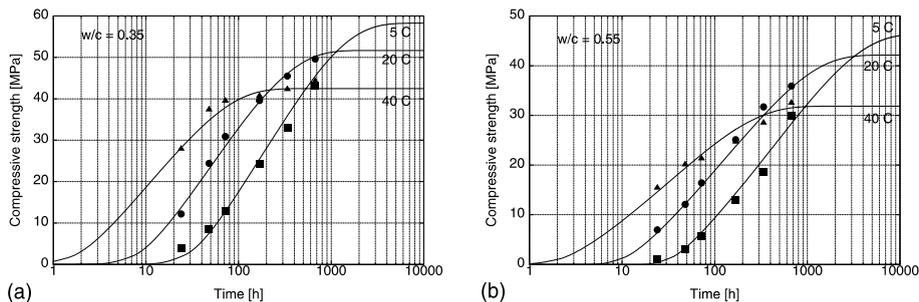


Fig. 4. Strength evolution for the isothermal Kim et al. tests.

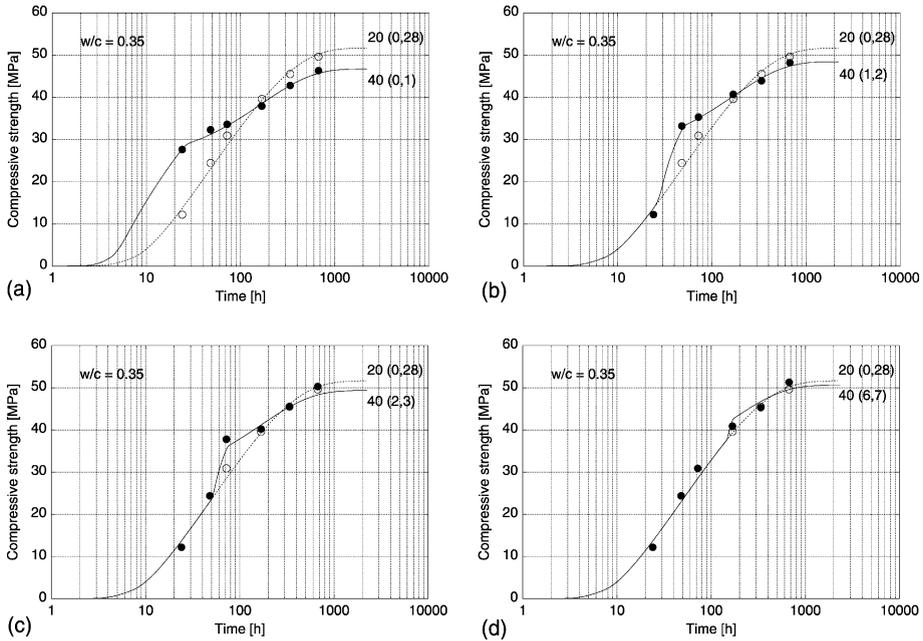


Fig. 5. Strength evolution for the $w/c = 0.35$ Kim et al. tests.

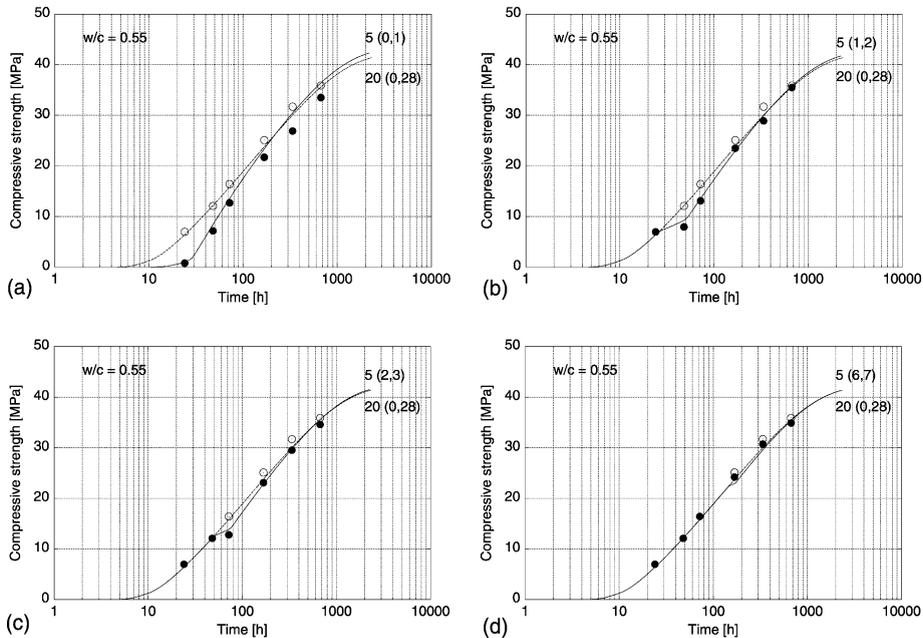


Fig. 6. Strength evolution for the $w/c = 0.55$ Kim et al. tests.

compressive strength evolution due to the elevation of the curing temperature above the 20 °C, or conversely, the retarding effect produced on concrete aging due to reducing the temperature to 5 °C, (ii) the progressive

attenuation of the influence of the one day thermal variations as such variations are produced on later ages, practically cancelling after the 6–7th day, irrespective to the w/c ratio and the stepwise thermal change.

4. Application to a concrete deck

Fig. 7 illustrates the configuration of the viaducts of the Öresund Link, between Denmark and Sweden. The viaduct is made up by two steel truss beams, on top of which a concrete deck is connected. By virtue of the extreme environmental conditions expected for the site of the Öresund Link—temperatures below $-20\text{ }^{\circ}\text{C}$, marine atmosphere, cycles of freeze-thaw—quite demanding casting and curing conditions were specified for the decks, which compelled the constructor (Dragados offshore) to build a curing warehouse for protecting the viaducts, inside of which the air temperature was kept around $20\text{ }^{\circ}\text{C}$ via an acclimatization system. In order to control the casting and the curing conditions several prototypes of the viaducts were monitored via thermal sensors, whose temperature measurements will be used to validate the described numerical model.

Fig. 8 depicts the left half of the viaduct concrete deck, as well as the four-noded FE mesh adopted for the discretization and the location of the four thermal sen-

sors (sensors 1, 6, 10 and 11) adopted for comparison of the experimental measurements with the numerical predictions. Computations started with the calibration of the hydration and aging model parameters, task that was undertaken on the basis of the concrete characterization performed in the laboratory that supported the constructor, and lead to the set of model parameters resumed in Table 3. The performed calibration is reproduced in Fig. 9a, which compares the ‘numerical’ and ‘experimental’ temperature rises for a concrete sample cured in adiabatic conditions, and in Fig. 9b, which depicts the numerical and experimental evolutions foreseen for the compressive strength during a $20\text{ }^{\circ}\text{C}$ isothermal test.

The thermal analysis performed for simulating the curing of the concrete deck is reported in Fig. 10, where the temperature evolutions foreseen by the numeric model are compared with the experimentally measured ones at sensors 1, 6, 10 and 11. For this numerical simulation two thermal contact conditions were considered along the deck boundaries: (i) one at the top,

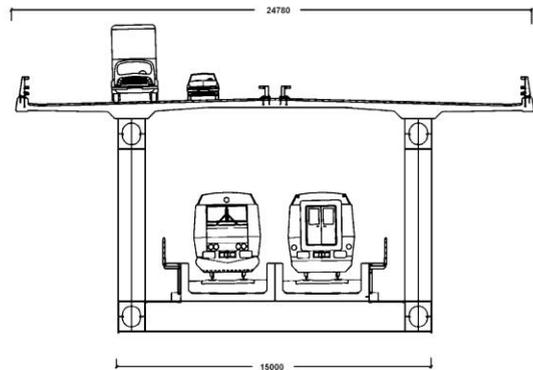
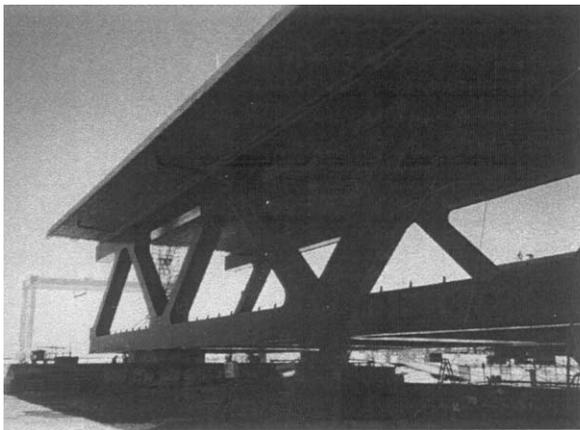


Fig. 7. Öresund Link viaduct.

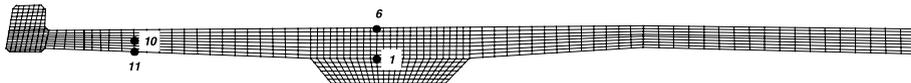


Fig. 8. Four-noded FE mesh and location of the thermal sensors.

Table 3
Material properties for the Öresund concrete deck

w/c	ζ_{∞}	k/n_0 (10^7 h^{-1})	\bar{n}	A_0/k (10^{-4})	E_a/R ($^{\circ}\text{K}$)	Q_{ζ} (10^8 J/m^3)	A	B	ζ_{set}	f_{∞} (MPa)	a
0.32	0.64	0.27	8.4	1.0	4000	1.95	2.19	0.16	0.1	53.4	0.0

$C = 1.87 \times 10^6\text{ J/m}^3\text{ }^{\circ}\text{C}$, $T_0 = 20\text{ }^{\circ}\text{C}$, $T_r = 20\text{ }^{\circ}\text{C}$.

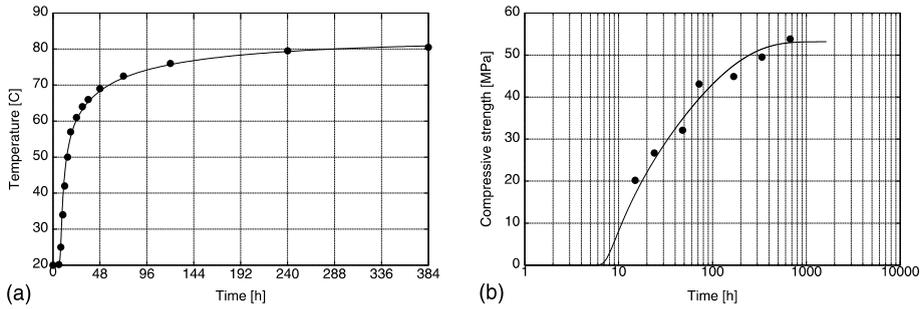


Fig. 9. Calibration tests: (a) temperatures on an adiabatic test; (b) compressive strength evolution on a 20 °C isothermal test.

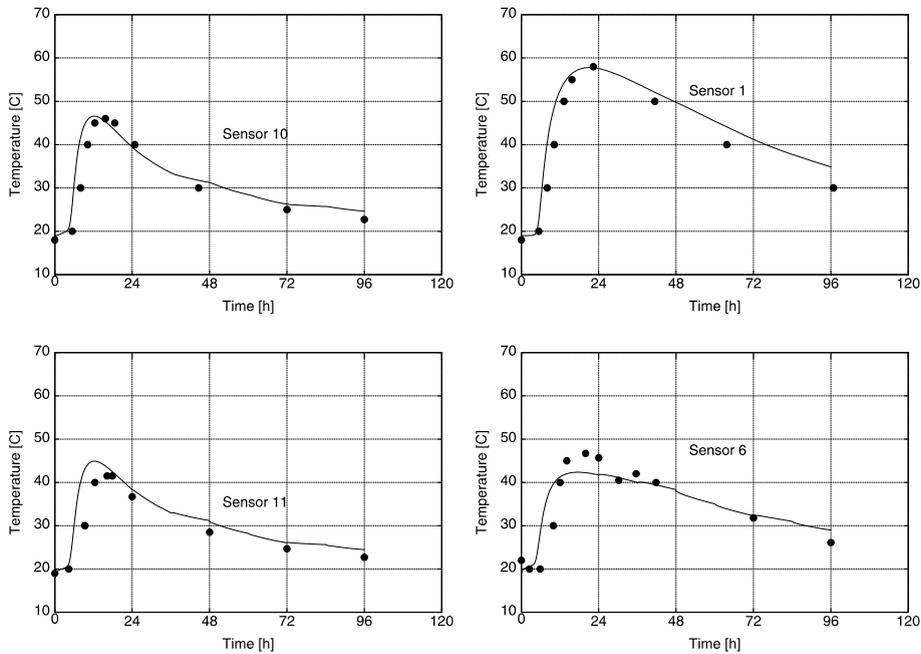


Fig. 10. Evolution of temperature during the curing of the concrete deck.

corresponding to the plastic sheet which recovered the concrete surface during the casting and curing process, and (ii) one at the bottom, representing the formwork conductivity properties (for further details see [11]).

As demonstrated in Fig. 10 the numeric model provides an adequate prediction of the overall concrete curing phenomenon, expressed through the reasonably accurate guessing of the temperature evolutions registered at the four sensors. It is remarked the fast temperature rise during the first day after casting, with a maximum temperature jump of about 40 °C at sensor 1, which is closer to adiabatic conditions by virtue of the largest deck thickness and of the internal placing of the sensor. Fig. 10 demonstrates that the proposed model is also able to predict the cooling phase which follows the initial heating process, a relevant feature since associated

to the contraction of the concrete deck, therefore associated to the undesirable generation of tensile stresses. Note that the heating and the cooling phases perform differently from one sensor to the other, yet reasonably captured by the numerical model, putting into evidence the ability of the latter in providing physically sound predictions of both the time and the spatial temperature distributions induced by the concrete curing.

5. Conclusions

With a format suitable for structural applications a model that accounts for the hydration phenomenon at a macroscopic level is proposed in this work. It calls for a normalised variable termed hydration degree, for which

an evolution law is presented. In order to cope with structural applications the thermal problem arising from the internal heat source provided by the hydration reaction needs to be solved, which is accomplished by adopting a FE spatial discretization and a time integration based on the explicit Euler scheme. Through comparison with suitable experimental tests performed for ordinary and high performance concretes accurate predictions for the hydration degree and the temperature evolutions are obtained during the curing process.

As for the evolution of the concrete compressive strength the novel concept of aging degree is introduced, accounting for both the hydration degree and the temperature influences on the aging of concrete during the curing. An important feature easily captured by the proposed aging model is the temperature influence on the final concrete strength, that is, at the end of the hydration reaction. Model capabilities are checked by comparing several experimentally observed strength evolutions at the early ages of concrete with the ones obtained numerically.

Suitability of the proposed numerical model for real structural applications is demonstrated by simulating the concrete curing reported for an experimentally instrumented viaduct deck. This application puts into evidence the ability of the proposed model to provide accurate predictions of the temperature evolutions registered on several thermal sensors installed on the deck.

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